

## Charge density measurement in $\text{MgH}_2$ by synchrotron X-ray diffraction

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### Abstract

Magnesium and its alloys are considered one of the most promising materials for reversible hydrogen storage. However, their high thermodynamic stability is unfavorable for dehydrogenation processes. In order to improve their properties, understanding of the bonding nature of Mg and H is essential. In this study, we reveal the precise experimental charge density distribution in  $\text{MgH}_2$ . X-ray powder diffraction data were obtained using the synchrotron radiation. The charge density distribution was calculated by the MEM (maximum entropy method)/Rietveld analysis. At room temperature, the charge density distribution around Mg is spherical, whereas the lower charge density distribution around H is non-spherical and slightly spread in the direction of the nearest neighbor Mg and H atoms. The number of electrons within the sphere around the Mg and the H atoms were estimated from the obtained distribution. As the result, Mg is almost fully ionized as  $\text{Mg}^{2+}$ , whereas hydrogen is very weakly ionized. The ionic charge of hydrogen is lower than the theoretical value. © 2003 Elsevier B.V. All rights reserved.

**Keywords:** Hydrogen storage materials; Magnesium hydride; X-ray diffraction; Charge density; Synchrotron radiation

### 1. Introduction

Hydrogen is considered to be one of leading candidates for clean energy sources in the future. For the practical use of hydrogen, development of hydrogen storage technology is required. For safe and efficient hydrogen storage, developments of new hydrogen storage alloys are currently being researched. The hydrogen storage capacity per unit weight of typical metal alloys is very low and not sufficient for use in a fuel cell vehicle. Therefore, alloys containing light elements such as Li and Mg, are focused on as high performance storage materials. Magnesium especially has a high storage capacity. For this reason, magnesium and its alloys are considered to be one of the most important candidates for reversible hydrogen storage material.

Another important property of a hydrogen storage material is its hydrogen desorption temperature. Unfortunately, magnesium hydrides are thermodynamically stable. The dehydrogenation of magnesium hydrides requires high temperatures (>550 K). Studies of nanometer-scale structure and nano-composites for magnesium and its

alloys were developed to lower the thermodynamic stability, thus increasing the dehydrogenation performance [1,2]. Furthermore, understanding the bonding nature of Mg and H is essential in order to improve its fundamental dehydrogenation performance. Several theoretical investigations on charge density distribution in  $\text{MgH}_2$  have been reported [3–6]. But, the bonding nature and ionic state of  $\text{MgH}_2$  have not been studied experimentally.

The position of hydrogen atoms in metal hydrides are usually determined by neutron diffraction analysis. Charge density distribution is typically investigated by X-ray diffraction analysis, but the diffraction intensity from hydrogen atoms in metal hydrides is very weak. So far, it has been difficult to determine the position and the charge density distribution of hydrogen by X-ray diffraction. In this study, we measured the precise X-ray diffraction intensity and analyzed the charge density distribution of hydrogen in metal hydrides using synchrotron radiation [7].

### 2. Experiment and analysis

The  $\text{MgH}_2$  sample was prepared by hydrogen activation treatment of pure magnesium powder (purity: 99%). The activation treatment was performed by the repetition of

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absorption under hydrogen pressure of 7 MPa and desorption by a rotary vacuum pump (5 Pa). These treatments were repeated five times at 573 K. The small size (sub-micron) powder was selected from the  $\text{MgH}_2$  sample by precipitation. The obtained fine powder was inserted into a glass capillary (diameter: 0.1 mm) for X-ray diffraction measurement.

X-ray powder diffraction data were obtained using synchrotron radiation at the beam-line BL02B2 in SPring-8. The X-ray diffraction experiment was carried out by use of the large Debye–Scherrer camera with imaging plate as detectors. Further details of this apparatus are given in Ref. [7]. The X-ray powder diffraction pattern was measured at room temperature. To fully use the dynamic range of the imaging plate, the exposure time of 53 min was used for this sample. The wavelength of incident X-ray was 0.696 Å. The X-ray diffraction intensities were obtained with a  $0.02^\circ$  step from  $5.0^\circ$  to  $73.0^\circ$  in  $2\theta$ , which corresponds to 0.585 Å resolution.

The experimental data were analyzed by the Rietveld method using a computer program RIETAN [8]. At first, X-ray scattering factors of neutral atoms (Mg and H) were used in Rietveld analysis. MEM (maximum entropy method) analysis revealed that Mg in  $\text{MgH}_2$  was in an ionic state. Therefore, the scattering values of  $\text{Mg}^{2+}$  ion and H atom were used in the final refined analysis.

The charge density distribution in a crystal of  $\text{MgH}_2$  was obtained by the MEM/Rietveld method. The MEM/Rietveld method, which is a combination of the MEM and the Rietveld refinement, is successfully employed in experimental investigations of charge density distribution in a crystal [9]. Details of the method are described in Ref. [10]. We applied this method to metal hydrides for the first time, and revealed experimentally, the bonding nature of  $\text{MgH}_2$ .

The MEM charge density analysis was carried out by

use of the observed structure factors of 103 reflections derived from the Rietveld analysis. The MEM calculation was carried out in  $48 \times 48 \times 32$  pixels per tetragonal lattice using the computer program ENIGMA [11].

### 3. Results and discussion

The X-ray powder diffraction sample was composed of three phases,  $\text{MgH}_2$ , Mg and MgO. By Rietveld analysis, the mass fractions for each phase were determined as 69.0, 27.0 and 4.0 wt% for  $\text{MgH}_2$ , Mg and MgO, respectively. The lattice parameter of  $\text{MgH}_2$  is  $a=4.5180(6)$ ,  $c=3.0211(4)$  Å, which is a little larger than that of  $\text{MgD}_2$  [12,13]. The weighted profile reliability factor,  $R_{\text{wp}}$ , and the reliability factor based on integrated intensities,  $R_1$ , of the Rietveld analysis were 2.9 and 1.7%, respectively. The fitting result of the Rietveld analysis is shown in Fig. 1. The inserted figure clearly shows the  $\{111\}$  reflection at  $18.2^\circ$  and the  $\{210\}$  reflection at  $19.8^\circ$ , which are attributed to the sub-lattice of only H atoms.

In MEM analysis, the reliability factor  $R$  was 1.0%, which is small enough to determine the charge densities of hydrogen and its bond nature. Fig. 2 illustrates the crystal structure of  $\text{MgH}_2$  (rutile type). The charge density distribution of (001) plane in the crystal of  $\text{MgH}_2$  is shown in Figs. 3 and 4. At room temperature, the charge density distribution around Mg is spherical, whereas the lower charge density distribution around H is non-spherical and slightly spread in the direction of the nearest neighbor Mg and H atoms.

The number of electrons within the Mg atom region (within the sphere of 0.95 Å radius) was estimated as 10.09e from the obtained charge density distribution. Similarly, the number of electrons within the H atom region (within the sphere of 1.00 Å radius) was estimated

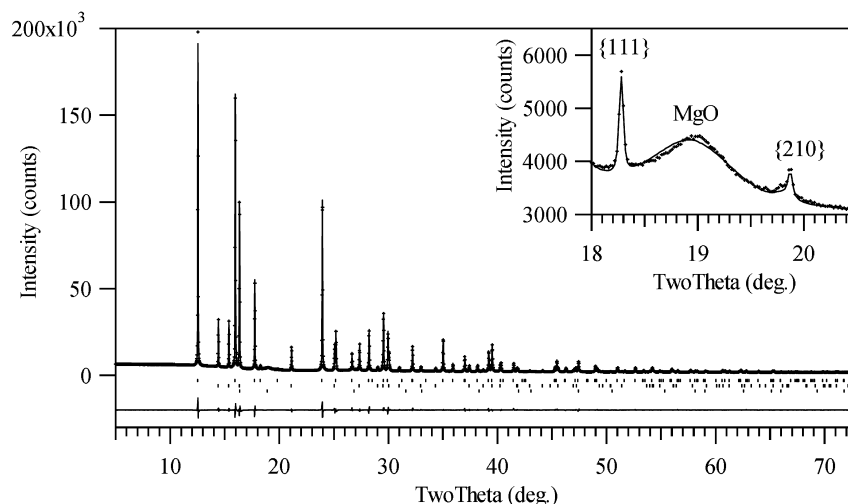


Fig. 1. Rietveld analysis pattern of the  $\text{MgH}_2$  sample at room temperature.  $\text{MgH}_2$ , Mg and MgO are included in this sample. The enlargement pattern of the region from  $18.0^\circ$  to  $20.5^\circ$  is shown in the inset.

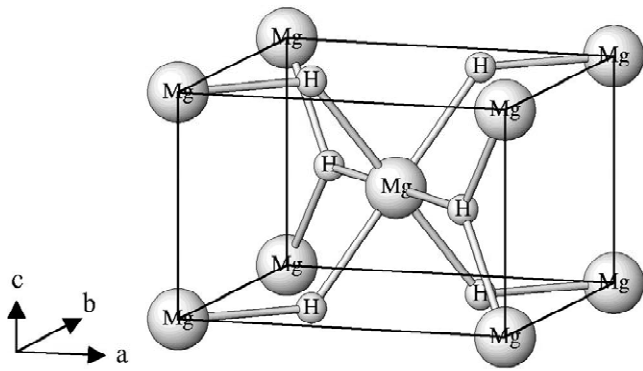


Fig. 2. The crystal structure of  $\text{MgH}_2$  (rutile type).

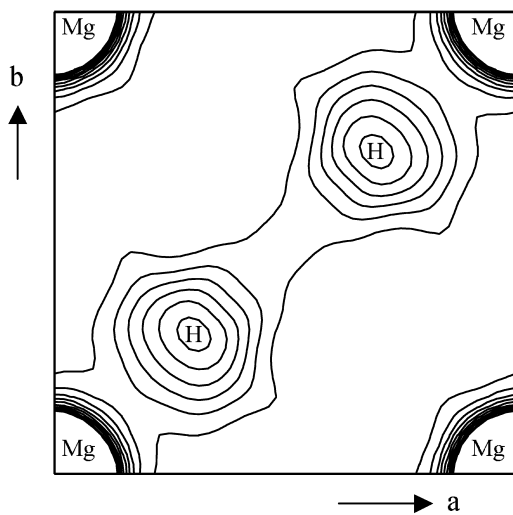


Fig. 3. The charge density distribution map in (001) plane of  $\text{MgH}_2$  at room temperature by MEM. The contour lines are drawn from 0.0 to 1.5 at  $0.15 \text{ e}/\text{\AA}^3$  intervals.

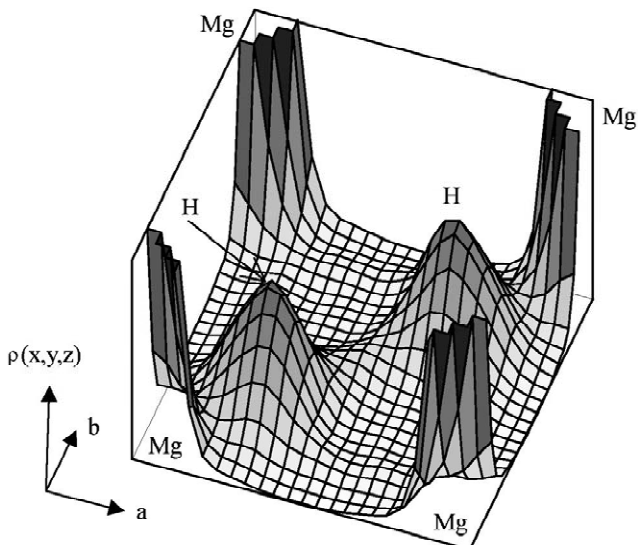


Fig. 4. The charge density distribution function in (001) plane of  $\text{MgH}_2$  at room temperature.

as 1.26e. These values indicate that ionic charge of Mg and H are represented as  $\text{Mg}^{1.91+}$  and  $\text{H}^{0.26-}$ . The results indicated that Mg is almost fully ionized as  $\text{Mg}^{2+}$ , whereas hydrogen is very weakly ionized. Experimentally, the ionic charge of hydrogen is lower than the theoretical values of  $\text{Mg}^{1.97+}$  and  $\text{H}^{0.48-}$  [4],  $\text{Mg}^{1.95+}$  and  $\text{H}^{0.60-}$  [5] or  $\text{Mg}^{1.886+}$  and  $\text{H}^{0.943-}$  [6]. More details about this experiment and further discussion are reported in Ref. [14].

#### 4. Conclusion

X-ray powder diffraction data for  $\text{MgH}_2$  were precisely obtained at room temperature using the synchrotron radiation. The electron density distribution in the crystal of  $\text{MgH}_2$  was obtained by the MEM/Rietveld analysis. The charge density distribution around Mg atom is spherical. The lower charge density distribution around H atom is non-spherical slightly spreading in the direction of the nearest neighboring Mg and H atoms. The numbers of electrons within the sphere around Mg and H atoms were estimated as 10.09e and 1.26e, respectively, from the obtained distribution. This means that Mg in  $\text{MgH}_2$  is fully ionized, but H atoms are in a weak ionic state. The ionic state and charge density distribution, which should be related to the stability of hydrogen in  $\text{MgH}_2$ , were revealed experimentally.

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